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Efficient CO₂ reduction under visible light: Synergistic effects of Cu nanoparticles and Ni single atoms

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ABSTRACT

Photocatalytic reduction of CO_2 is a considerable method to alleviate global warming and energy shortage. In this work, CoAl-LDH/ Ni_1Cu_{NP} nitrogen-doped carbon composites (CA/Ni_1Cu_{NP} N-C) can efficiently reduce CO_2 into CO, CH_4 , and C_2H_6 under visible light. In situ characterization techniques and theoretical calculations were employed to investigate the CO_2 evolution pathway and photocatalytic mechanism. The synergistic interaction of CU nanoparticles and Ni single-atom sites accelerated the photogenerated charge separation and facilitated the adsorption properties of key intermediates. As a result, the yields of CH_4 and C_2H_6 on CA/Ni_1Cu_{NP} N-C were achieved up to 35.245 and 25.328 μ mol g⁻¹ h⁻¹ with electron selectivity of 40.24% and 50.61%, respectively. This work is instructive for using metal nanoparticles to optimize single-atom catalysts to promote their catalytic activity.

1. Introduction

The massive carbon dioxide (CO2) emissions have led to drastic climate changes and frequent extreme natural disasters. Reducing CO2 into value-added fuels and chemicals by artificial photosynthesis is a promising solution to environmental problems and energy crises. [1,2] In recent years, some semiconductor materials and reaction systems with superior photocatalytic properties have been developed, and significant progress has been made in energy conversion efficiency, operational stability, and product selectivity. [3-5] Photocatalytic reduction of CO2 into CO and HCOOH is kinetically favorable because their generation requires the transfer of only 2 electrons. However, converting CO2 into products with higher economic value is very difficult. For example, the generation of CH₄ is thermodynamically favorable, but its evolution involves the adsorption of *CO intermediates and 8-electron transfer, which are kinetically challenging. On this basis, the generation of C₂H₆ further involves the rate-determining step (RDS) of C-C coupling and more hydrogenation reactions. [6-8] Therefore, designing novel photocatalysts with high carrier separation efficiency, moderate intermediate adsorption capacity, and sufficient active sites is of great significance for generating high-value products.

Narrowing the scale of conventional bulk materials to nanoparticles,

clusters, or even single atoms can exponentially increase the number of active sites. [9] In particular, single-atom catalysts theoretically possess 100% utilization since they are well dispersed on the atomic scale. [10] In addition, isolated single-atom sites have unique electronic structures and coordination environments, which are favorable for CO2 catalytic reduction with complex reaction pathways. [11,12] The zeolite imidazole framework-8 (ZIF-8) formed by the coordination of Zn²⁺ and 2-methylimidazole is one of the classical metal-organic frameworks (MOFs) with a well-defined coordination pattern and dispersed pore structure. [13] Therefore, ZIF-8 is an ideal precursor for designing controllable morphology and functionalized catalysts. Substituting a portion of Zn²⁺ based on the metal ion exchange strategy can obtain atomically isolated metal-doped species (M), which is a mature technology for anchoring metal single atoms on ZIF-8-derived nitrogen-doped carbon (N-C). [14] Since N atoms tend to establish strong electron-metal-carrier interactions (EMCIs) with metal species, it can effectively anchor metal atoms and maintain their atomic dispersion at high temperatures. [15] In addition, the ZIF-8-derived single-atom catalyst inherits its special pore structure and possesses superior electrical conductivity, which is favorable for the adsorption of reactants/intermediates and rapid charge migration, and thus it has been widely studied in energy and environmental catalysis. [16-18]

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However, the typical M-N $_4$ coordination has a symmetric electronic structure, which is not conducive to efficient carrier separation and C-C coupling. [19–21] The microenvironment of the active center can be modulated by doping elements, adjusting the coordination mode, and designing double single-atom sites, thus facilitating the conversion of ${\rm CO}_2$ to ${\rm CH}_4$ and ${\rm C}_2{\rm H}_6$. [22–24] Recently, several reports have found that metal nanoparticles coexisting with single atoms exhibit better catalytic performance than their individual components. [25–28] The introduction of metal nanoparticles creates new electron-rich centers, which can significantly change the charge distribution of the original single-atom sites. The establishment of the electron density difference can promote the separation of carriers and the aggregation of electrons. Meanwhile, the asymmetric charge densities around single-atom sites facilitates the coupling of *CO intermediates to promote the generation of multi-carbon products [20,29].

However, direct use of metal nanoparticle-single atom synergistic system based on N-C for photocatalytic CO2 reduction is undesirable, due to the limitation of available photogenerated carriers. Adopting mature photocatalytic materials as photogenerated carrier sources and constructing heterojunctions with N-C can provide sufficient electrons to promote the catalytic reduction of CO₂. Layered double hydroxides (LDHs) with unique structures and good adsorption properties have been widely studied in non-homogeneous photocatalytic reaction systems. [30-32] Particularly, Co²⁺ in CoAl-LDH can accelerate the dissociation of H₂O thus promoting the catalytic reduction of CO₂. [33] However, CoAl-LDH-based photocatalysts tend to exhibit poor energy conversion efficiency due to the rapid recombination of photogenerated carriers and the lack of active sites. [34,35] CoAl-LDH and ZIF-8-derived N-C have adaptable energy-band structures, and hybridizing them can promote the transfer of photogenerated carriers to N-C. Meanwhile, the abundant active sites on N-C can efficiently reduce CO2 under sufficient electron flow.

In this work, we anchored Cu nanoparticles and Ni single-atom sites on N-C (Ni₁Cu_{NP} N-C), which were then hybridized with CoAl-LDH (CA) by ultrasonic self-assembly method to obtain $CoAl-LDH/Ni_1Cu_{NP}$ N-C (CA/Ni₁Cu_{NP} N-C) composites for photocatalytic CO₂ reduction. The yields of CH₄, CO, and C₂H₆ were 35.245, 32.067, and 25.328 μmol g h⁻¹, respectively, and their electron selectivity were 40.24%, 9.15%, and 50.61%, respectively. The product generation pathways, charge transfer mechanisms, and synergistic effects between Cu nanoparticles and Ni single-atom sites were investigated by in situ characterization techniques and DFT calculations. The atomically dispersed Ni singleatom sites provide sufficient active sites for the photocatalytic reaction. The loading of Cu nanoparticles establishes electron-rich centers that accelerate charge separation and transport. In addition, it also optimizes the electronic structure of the Ni center, which promotes the adsorption properties of the key intermediates and the C-C coupling step.

2. Results and discussion

2.1. Synthesis and characterization

Based on the conventional ZIF-8 preparation process, a certain amount of Cu^{2+} and Ni^{2+} was substituted for Zn^{2+} (the molar ratio of $(\text{Cu}^{2+}+\text{Ni}^{2+})/\text{Zn}^{2+}$ is 5%) to obtain NiCu@ZIF-8. Subsequently, ZIF-8-derived N-C with Cu nanoparticles and Ni single-atom sites (Ni₁Cu_{NP} N-C) was obtained by one-step pyrolysis under a nitrogen atmosphere. CoAl-LDH (CA) was prepared by a hydrothermal method, and the CA/Ni₁Cu_{NP} N-C composite catalyst was synthesized via ultrasonic self-assembly. For comparison tests, ZIF-8, Cu@ZIF-8, and Ni@ZIF-8 were synthesized and the corresponding N-C-based materials (M N-C) were obtained in the temperature range of 800–950 °C. Finally, a series of composite catalysts were synthesized by controlling the ratio of CA and M N-C (CA/M N-C).

The crystal structure of the as-prepared sample was studied by

powder X-ray diffraction (XRD). As shown in Fig. 1a, the diffraction peaks of pristine ZIF-8 at 10.46°, 12.8°, 14.78°, 16.52°, 18.1°, 22.22°, 24.58°, and 26.76° correspond to its (002), (112), (022), (013), (222), (114), (233), and (134) crystal faces, respectively. [36,37] These typical crystal surfaces verify the successful preparation of ZIF-8. Cu@ZIF-8, Ni@ZIF-8, and CuNi@ZIF-8 also exhibit diffraction peaks consistent with those of ZIF-8, which implies that the stabilized structure of ZIF-8 is preserved during the metal-ion exchange process. However, the diffraction peaks associated with the elements Cu and Ni are not detected due to their excessively low content and atomic dispersion. The optimal N-C, Cu_{NP} N-C, and Ni_1Cu_{NP} N-C were obtained by calcination under nitrogen atmosphere at 950 °C. As shown in Fig. 1b, the broad diffraction peaks near 23.34° and 43.54° correspond to the (002) and (101) facets of amorphous carbon. [38] For CuNP N-C, three distinct characteristic peaks at 13.14°, 50.32°, and 74.06° correspond to the (111), (200), and (220) crystal faces of Cu⁰, respectively. [39] The diffraction information of Ni_1Cu_{NP} N-C is similar to that of Cu_{NP} N-C. In particular, no Ni signal is detected on Ni₁Cu_{NP} N-C, which implies the atomic dispersion of Ni. The Ni₁Cu_{NP} N-C obtained at 800, 850, and 900 °C also exhibit similar diffraction information. As shown in Fig. S1, the characteristic peaks of Cu species are gradually enhanced with increasing temperature, which implies that high temperature is more favorable for Cu aggregation. The characteristic (003), (006), (009), (012), (018), (110), and (113) crystal faces of CA can be observed in its XRD pattern. [40,41] In addition, the diffraction peaks of each monomer are successfully detected on CA/Ni₁Cu_{NP} N-C (Fig. 1c), which means that the composite catalyst is successfully synthesized by ultrasonic self-assembly method.

The vibrational properties of N-C, Cu_{NP} N-C, Ni₁ N-C, and Ni₁Cu_{NP} N-C were investigated via Raman spectroscopy. As shown in Fig. 1d, all the samples exhibit distinct broad peaks near 1350 and 1580 cm⁻¹, which correspond to the C-atom lattice defects (D-band) and the phase crystalline graphite (G-band), respectively. The peak intensity ratio of the D and G bands (I_D/I_G) can evaluate the degree of graphitization. It is obvious that doping metal is conducive to improving the crystal quality of N-C, thus obtaining better photoelectron transport efficiency. [42,43] The specific surface area and pore size of catalysts play an essential role in the mass transfer of reactants and adsorption of key intermediates during heterogeneous reaction. [44] Therefore, the pore structure of the as-prepared samples was investigated by nitrogen absorption measurements (Fig. S2 and Table S1). Compared to CA (55.435 m² g⁻¹), CuNi@ZIF-8 has a larger Brunauer-Emmett-Teller (BET) surface area (1113.903 m² g⁻¹), which is attributed to the unique mesoporous structure of ZIF-8. After pyrolysis, the specific surface area of Ni₁Cu_{NP} N-C decreases dramatically (479.938 m² g⁻¹) due to the collapse of the carbon structure and the formation of defects. [45] As the low content of CA, CA/Ni₁Cu_{NP} N-C possesses a similar specific surface area $(427.907 \text{ m}^2 \text{ g}^{-1})$ and pore size as Ni₁Cu_{NP} N-C.

The microstructure of the as-prepared samples was investigated by scanning electron microscopy (SEM), transmission electron microscopy (TEM), high-resolution transmission electron microscopy (HR-TEM), energy dispersive spectrometer (EDS), and high-angle annular darkfield scanning transmission electron microscopy (HAADF-STEM). As shown in Fig. 2a, CA exhibits a nanoflower-like shape formed by the aggregation of 2D nanosheets. The structure of CA is relatively loose, and some smaller nanosheets are attached to its surface (Fig. S3a and Fig. 2b). N-C transformed from pure ZIF-8 exhibits a rhombic dodecahedral structure. Due to the reduction and evaporation of Zn and the disappearance of organic ligands at high temperatures, N-C presents a rough surface. [46] In particular, some N-C particles directly expose their internal coral-like skeleton and no longer have an intact surface (Fig. S3b and Fig. S4). After being loaded with Cu and Ni, a few uniformly distributed nanoparticles are observed on the N-C (Fig. 3a and Fig. S3c), and the EDS results confirmed that they are Cu nanoparticles formed by the reduction and aggregation of Cu²⁺ at high temperature (Fig. S5). Correspondingly, the Ni element is spatially uniformly dispersed, which implies that Ni

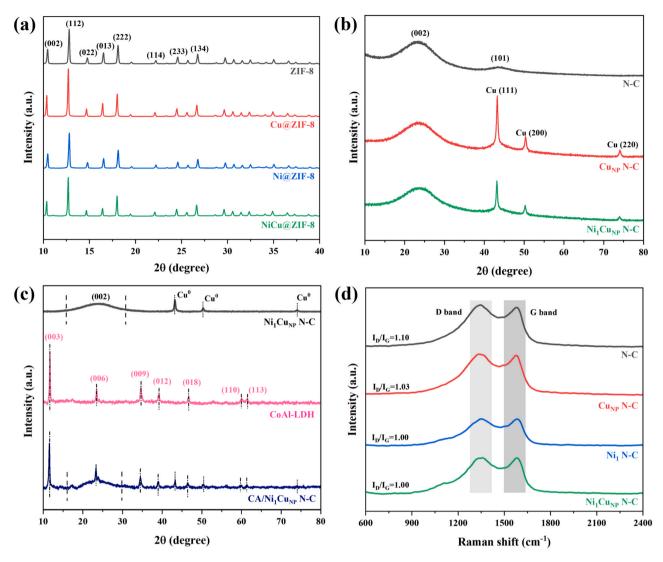


Fig. 1. The XRD patterns of (a) ZIF-8, Cu@ZIF-8, Cu@ZIF-8, Cu@ZIF-8, Cu_{NP} N-C, Cu_{NP} N-C, Cu_{NP} N-C, Cu_{NP} N-C. (c) Cu_{NP} N-C. (c) Cu_{NP} N-C. (c) Cu_{NP} N-C. Cu_{NP}

was successfully embedded into N-C as single-atom sites. The carbon products evolved from CuNi@ZIF-8 show a more regular dodecahedral structure (Fig. 2c-d). The TEM images of Ni₁Cu_{NP} N-C also show that some regular cyclic graphitized carbon is formed at the edge of the N-C. The HR-TEM (Fig. 3c) and HAADF-STEM (Fig. S6) images of Ni₁Cu_{NP} N-C show that the lattice gap of the nanoparticles is about 0.208 nm, which corresponds to the (111) crystal faces of Cu⁰. [41] Since metallic Cu can promote carbon evolution, the Cu nanoparticles are uniformly encapsulated by graphitized carbon (layer spacing is approximately 0.34 nm) to form a unique core-shell structure. The carbon layer with superior electrical conductivity can promote the rapid migration of electrons to active sites, which is favorable for the CO2 reduction reaction with multi-electron participation. [47,48] For CA/Ni₁Cu_{NP} N-C composite catalysts, CA is dispersed into nanosheets and rosette-like structures after sonication and distributed in the interstitials of Ni₁Cu_{NP} N-C (Fig. 2e-f and Fig. S3d). Its EDS mapping photos are similar to those of Ni₁Cu_{NP} N-C, which suggests that sonication did not change its surface metal dispersion state (Fig. S7). In addition, HAADF-STEM image analysis further confirmed the atomic dispersion of Ni on CA/Ni₁Cu_{NP} N-C, and these isolated Ni sites are highlighted by red circles (Fig. 3d).

The surface element composition and chemical state of the samples were determined by X-ray photoelectron spectroscopy (XPS). The survey

XPS spectra of CA, Ni_1Cu_{NP} N-C, and CA/ Ni_1Cu_{NP} N-C are shown in Fig. S8. Due to the low content of Ni and Cu, no obvious characteristic peaks belonging to them are detected in the survey spectra. The spinorbit double peaks belonging to Co 2p3/2 (781.22 eV) and Co 2p1/2 (797.36 eV) and the corresponding satellite peaks located at 787.24 and 804.04 eV can be detected in the pristine CA HR-XPS Co 2p spectrum (Fig. S9a). The characteristic peak located near 74.39 eV in the Al 2p spectrum (Fig. S9b) indicates the presence of Al³⁺. [41] The HR-XPS spectra of Co, Al, C, N, Ni, and Cu elements in CA/Ni₁Cu_{NP} N-C are shown in Fig. 4. Compared with the pristine CA, the positions of the corresponding characteristic peaks of Co 2p and Al 2p are shifted toward the direction of high binding energy, which indicates that the electron cloud density of this component decreases and electrons are more inclined to migrate to Ni₁Cu_{NP} N-C. [40] In the HR-XPS C 1 s spectrum of CA/Ni₁Cu_{NP} N-C, the peaks located at 284.77, 285.87, and 288.69 eV correspond to C-C, C=N, and N-C bonds, respectively. [49] The peaks located at 398.62, 399.36, 400.85, 402.09, and 404.62 eV belong to pyridinic N, Ni-N, pyrrolic N, graphitic N, and oxidized N, respectively, which is consistent with the pristine Ni₁Cu_{NP} N-C (Fig. S10). [50,51] Functionalized pyridine N can act as a coordination center to anchor metal single atoms. In addition, the presence of Ni-N bonds implies the existence of isolated Ni single atoms and coordination with N. [52] The characteristic peak belonging to Ni 2p3/2 (854.92 eV) is between Ni⁰

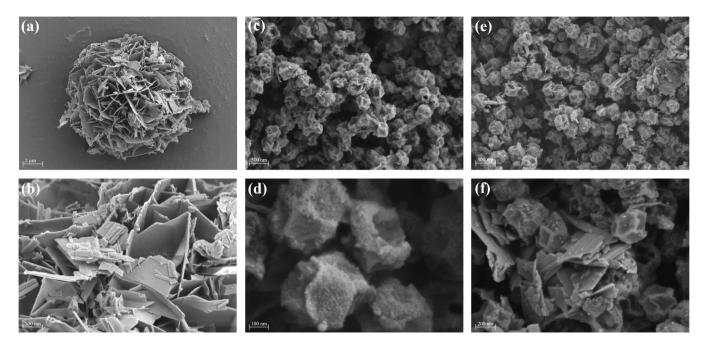
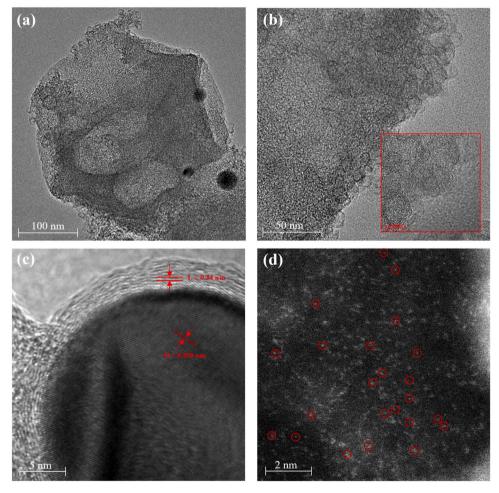


Fig. 2. The SEM images of (a-b) CoAl-LDH, (c-d) Ni_1Cu_{NP} N-C, and (e-f) CA/ Ni_1Cu_{NP} N-C.



 $\textbf{Fig. 3.} \ \ \text{The TEM image of (a-b)} \ \ Ni_1Cu_{NP} \ \ \text{N-C. (c)} \ \ \text{The HR-TEM image of Ni}_1Cu_{NP} \ \ \text{N-C. (d)} \ \ \text{The HAADF-STEM image of CA/Ni}_1Cu_{NP} \ \ \text{N-C.}$

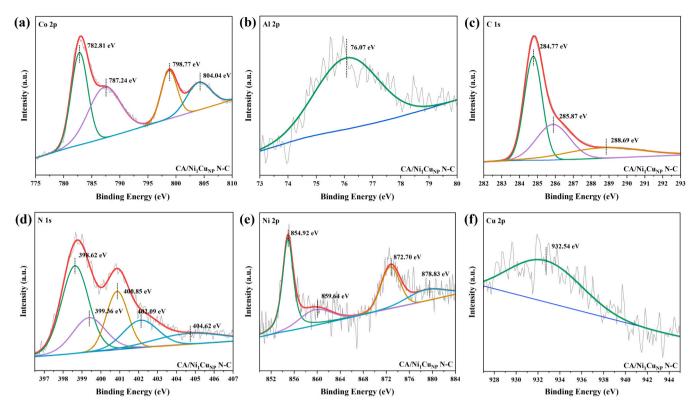


Fig. 4. The (a) Co 2p, (b) Al 2p, (c) C 1 s, (d) N 1 s, (e) Ni 2p, and (f) Cu 2p HR-XPS images of CA/Ni₁Cu_{NP} N-C.

(852.9 eV) and Ni $^{2+}$ (856.0 eV), which suggests that the valence state of Ni in the CA/Ni $_1$ Cu_{NP} N-C is between 0 and + 2. [53] As shown in Fig. 4f, the single peak detected near 932.54 eV belongs to Cu 0 2p3/2. [54] No

signals regarding ${\rm Cu}^+$ and ${\rm Cu}^{2+}$ are detected indicating that elemental ${\rm Cu}$ is mainly present in the form of ${\rm Cu}$ nanoparticles.

The electronic structure and coordination environment of Ni in CA/

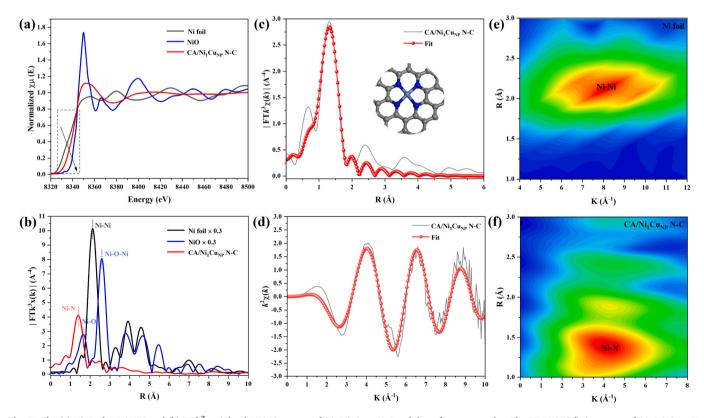


Fig. 5. The (a) Ni K-edge XANES and (b) FT k^3 -weighted EXAFS spectra of CA/Ni₁Cu_{NP} N-C and the reference samples. The FT-EXAFS fitting curve of CA/Ni₁Cu_{NP} N-C in (c) R-space (inset shows the proposed NiN₄ coordination environment) and (d) k-space. The WT-EXAFS signals of (e) Ni foil and (f) CA/Ni₁Cu_{NP} N-C.

Ni₁Cu_{NP} N-C were further confirmed by Ni K-edge X-ray absorption near edge structure (XANES) and extended X-ray absorption fine structure (EXAFS) characterization. As shown in Fig. 5a, the energy absorption edge of CA/Ni₁Cu_{NP} N-C is located between the standard Ni foil and NiO, which indicates that the valence state of Ni in CA/Ni₁Cu_{NP} N-C is between 0 and +2, which is consistent with the analysis results of XPS. As shown in Fig. 5b, no peaks associated with the Ni-Ni bond (2.12 Å), Ni-O-Ni bond (2.58 Å), and Ni-O bond (1.66 Å) are detected in the Fouriertransformed (FT) Ni k3-weighted EXAFS spectra, which indicates the absence of Ni nanoparticles and Ni oxides. Correspondingly, a characteristic peak associated with the scattering path of the Ni-N bond (1.41 Å) is detected in CA/Ni₁Cu_{NP} N-C, which suggests that the Ni in CA/Ni₁Cu_{NP} N-C is mainly in the Ni-N_X configuration similar to nickel phthalocyanine (Ni pc). [37] In addition, the Wavelet transform (WT) Ni K-edge EXAFS results (Fig. 5e-f) show that the intensity maximum of CA/Ni₁Cu_{NP} N-C in k-space is located at 4.07 Å⁻¹ without a correlation signal about the Ni-Ni coordination (9.08 Å^{-1}) in Ni foils, which further confirms that the atomic dispersion of Ni. Based on the EXAFS fitting results, the coordination number X was determined to be about 3.6, and it can be assumed that the Ni atoms in the CA/Ni₁Cu_{NP} N-C are coordinated to four N atoms and have a bond length of about 1.86 Å (Fig. 5c-d and Table S2).

2.2. Optical and photoelectrochemical properties

The light-absorbing ability of the photocatalyst is an essential index for evaluating its performance. As shown in Fig. 6a, CA has two distinct absorption bands in the UV and visible regions. [41] The black N-C, Cu_{NP} N-C, Ni_1 N-C, and Ni_1Cu_{NP} N-C exhibit superior light absorption in the UV and visible ranges, with an absorption edge of about 700 nm (Fig. S11). It is noteworthy that the light-absorbing ability of Cu_{NP} N-C and Ni_1Cu_{NP} N-C in the near-infrared region (NIR) is also significantly enhanced compared to N-C and Ni $_1$ N-C, which may be attributed to the localized surface plasmon resonance (LSPR) effect induced by the Cu nanoparticles. The CA/Cu_{NP} N-C, CA/Ni $_1$ N-C, and CA/Ni $_1Cu_{NP}$ N-C composites exhibit similar light-absorbing properties as those of the

corresponding M N-C-based materials due to the lower content of CA. As shown in Fig. 6b, the band gap (Eg) energies of CA, Ni_1Cu_{NP} N-C, and CA/Ni_1Cu_{NP} N-C are calculated as 3.41, 2.42, and 2.49 eV, respectively.

Electrochemical impedance spectroscopy (EIS) and transient photocurrent response (TPR) measurements were used to further investigate the charge transfer resistance and photocarrier separation efficiency of the prepared samples. The radius of the arc in the Nyquist diagram directly reflects the charge transfer resistance (Rct) of the samples, and it is clear that the composite CA/Ni₁Cu_{NP} N-C has the smallest Rct (Fig. 6e). By further fitting the test results, the R_{ct} of CA, CA/Cu_{NP} N-C, CA/Ni $_1$ N-C, and CA/Ni $_1$ Cu $_{NP}$ N-C are 4169, 464.5, 250.6, and 42.89 Ω , respectively (Table S3). As shown in Fig. 6d, the original CA exhibits a small photocurrent density, meaning that many carriers recombine during migration. Correspondingly, the transient photocurrents of CA/ Cu_{NP} N-C, CA/Ni₁ N-C, and CA/Ni₁Cu_{NP} N-C composites are significantly increased, which means that the hybrid of M N-C based materials and CA can substantially improve the photogenerated charge transport and separation mechanism. In the photoluminescence (PL) spectrum (Fig. 6e), CA exhibits the strongest peak near 420 nm, which is a fluorescence effect caused by the large number of recombination carriers during the transition. The fluorescence intensities of the composite samples are all reduced and the CA/Ni₁Cu_{NP} N-C has a minor fluorescence peak, further confirming its superior carrier separation efficiency. According to the time-resolved photoluminescence (TRPL) attenuation spectrum, the average carrier lifetime (Table S4) of CA, CA/Cu_{NP} N-C, CA/Ni₁ N-C, and CA/Ni₁Cu_{NP} N-C are 27.18, 29.99, 33.37, and 38.91 ns, respectively. The optimal carrier separation and transport properties of CA/Ni₁Cu_{NP} N-C may be due to the asymmetric electron density formed on the carbon substrate by the interaction of Ni_1 and Cu_{NP} . [15,55] Therefore, CA/Ni_1Cu_{NP} N-C can produce more valuable carriers to participate in the CO2 reduction reaction, promoting the formation of long carbon chain products involving multiple electron transfer.

2.3. Photocatalytic CO₂ reduction performance

A series of Ni₁Cu_{NP} N-C were prepared in the range of 800-950 °C

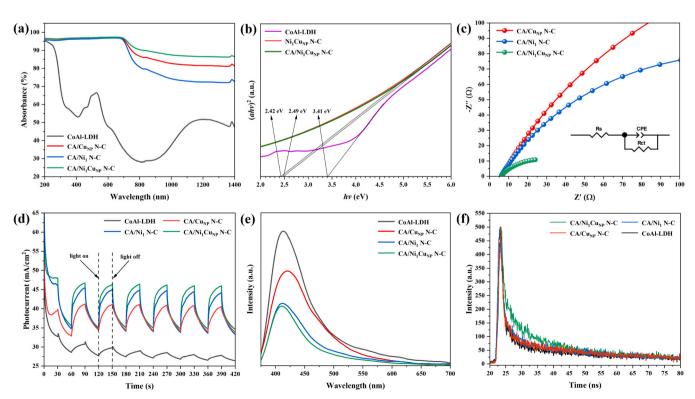


Fig. 6. The (a) UV-Vis-NIR absorption spectra, (b) Tauc plots, (c) EIS, (d) TPR, (e) PL, and (f) TRPL tests of as-prepared samples.

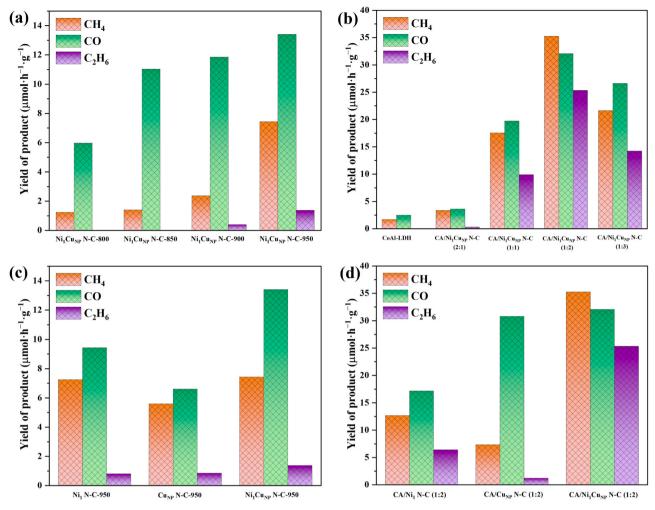


Fig. 7. The photocatalytic CO₂ reduction performance of as-prepared samples under visible light.

under the N₂ atmosphere to investigate the effect of calcination temperature on their photocatalytic performance. As shown in Fig. 7a, the photocatalytic performance of Ni₁Cu_{NP} N-C is enhanced with increasing calcination temperature. In particular, Ni₁Cu_{NP} N-C-900 and Ni₁Cu_{NP} N-C-950 can reduce CO₂ into C₂H₆. Higher temperatures can activate electron-metal-supporter interactions (EMSIs), when anchoring metal single atoms on N-C derived from ZIF-8 based on ion exchange strategies. This strong interaction is conducive to the coordination of metal ions and N sites. [15] In addition, the higher calcination temperature can ensure that the Zn species in the original ZIF-8 are fully evaporated, thus obtaining the target coordination pattern. Ni₁Cu_{NP} N-C-950 and CA were hybridized in different proportions by ultrasonic self-assembly method. Compared with the pristine CA, the photocatalytic activities of several CA/Ni₁Cu_{NP} N-C composites are enhanced. As shown in Fig. 7b, the yields of CH₄, CO, and C₂H₆ are increased with the percentage of Ni₁Cu_{NP} N-C since it can provide more active sites. Moreover, the selectivity of C2H6 has also been significantly improved. Under visible light, the yields of CH₄, CO, and C₂H₆ on CA/Ni₁Cu_{NP} N-C (1:2) are 35.245, 32.067, and 25.328 μ mol g⁻¹ h⁻¹, respectively, and the electron selectivity of C₂H₆ is about 50.61%. It can be observed that the photocatalytic performance of the composites decreases significantly when the ratio of Ni₁Cu_{NP} N-C is continuously increased. Although the black Ni₁Cu_{NP} N-C can improve the light absorption and active site of the composite photocatalysts, most of the incident light is not converted into usable photogenerated carriers, but released into the surrounding environment as heat energy. Therefore, the lower content of CA leads to a decrease in the actual photogenerated carrier, which is the main

reason for the decrease in photocatalytic activity.

Then, we evaluated the photocatalytic CO₂ reduction performance of Ni₁ N-C-950, Cu_{NP} N-C-950, and Ni₁Cu_{NP} N-C-950 under visible light to further investigate the synergistic effect of Ni single-atom sites and Cu nanoparticles, As shown in Fig. 7c, Ni₁Cu_{NP} N-C-950 exhibit the best photocatalytic activity, which implies that the interaction of Ni single atoms and Cu nanoparticles can promote CO₂ reduction. On CA/Cu_{NP} N-C, CO was the main gas-phase product due to the poor adsorption characteristics of the CuNP sites for *CO, so that *CO was easily desorbed on the catalyst surface. Correspondingly, CA/Ni₁ N-C has higher CH₄ and C₂H₆ yields while the CO yield is significantly decreased. This implies that the dispersed Ni single-atom sites have better adsorption properties for *CO intermediates, thus favoring the subsequent hydrogenation reaction to generate multi-electron transfer products. In particular, CA/Ni₁Cu_{NP} N-C show the best catalytic performance, which is attributed to the synergistic effect of Cu nanoparticles and Ni single atoms. The loading of Cu nanoparticles can establish electron-rich centers near the Ni single-atom sites to promote carrier separation and transport, thus obtaining sufficient electron sources to promote the CO₂ catalytic reduction reaction. On the other hand, establishing asymmetric Ni-N₄ active sites is conducive to C-C coupling, thus enabling the generation of C₂ products. Therefore, this synergistic effect can optimize the photogenerated carrier separation mechanism and promote C-C coupling, thus exhibiting enhanced photocatalytic activity and considerable CH₄ and C₂H₆ vields.

The isotope labeling tests were performed using $^{13}\text{CO}_2$ under the same test conditions, and the products were analyzed by gas

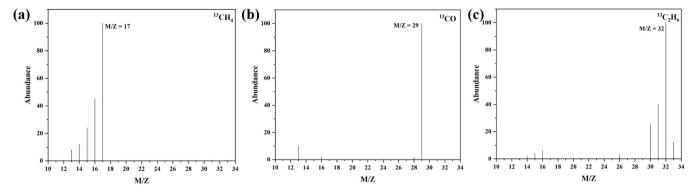


Fig. 8. GC-MS analysis for photocatalytic ¹³CO₂ reduction.

chromatography-mass (GC-MS) spectrometry. As shown in Fig. 8, the peaks located at M/Z=17, 29, and 32 corresponded to $^{13}\mathrm{CH_4}$, $^{13}\mathrm{CO}$, and $^{13}\mathrm{C_2H_6}$, respectively, which indicated that the products were all derived from $\mathrm{CO_2}$ rather than carbon pollution. The durability of the photocatalyst is a vital evaluation index for its application value. As shown in Fig. S12, the CH₄, CO, and $\mathrm{C_2H_6}$ all exhibit stable generation rates under 10 h of continuous irradiation. There is no significant decrease in the photocatalytic activity of the best sample CA/Ni₁Cu_{NP} N-C after three cycle tests (30 h). The SEM image (Fig. S13) and XPS pattern (Fig. S14) of the CA/Ni₁Cu_{NP} N-C sample after cyclic testing further confirm its structural stability.

2.4. Mechanism of photocatalytic CO₂ reduction

The energy band structures (Fig. S15) of CA and N-C were determined based on the Mott-Schottky (Fig. S16) and Tauc plots. Under irradiation, the electrons located on the conduction band (CB) of CA will be transferred to the CB of N-C by the internal potential, and then converge to the Cu nanoparticles and Ni single-atom sites to participate in the reduction reaction. Correspondingly, holes located in the valence band (VB) of CoAl-LDH will transfer to the VB of N-C and finally be consumed by the sacrificial agent (TEOA).

The CO₂-temperature programmed desorption (CO₂-TPD) tests up to 500 °C were carried out to further investigate the chemical adsorption properties of the prepared materials for CO₂. As shown in Fig. S17, the best sample CA/Ni₁Cu_{NP} N-C exhibits more obvious desorption peaks than CA/Cu_{NP} N-C at 127.3, 338.5, and 428.3 °C, which directly proves that the introduction of Ni single atomic sites could obtain enhanced chemisorption properties of CO2. In addition, a slight shift of the moderately alkaline sites of CA/Ni₁Cu_{NP} N-C to higher temperatures also indicates the presence of more adsorption sites. For inferring the CO₂ evolution pathway, in situ Fourier transform infrared spectroscopy (FT-IR) was used to determine the type of intermediates adsorbed on the catalyst surface during the catalytic reduction process. As shown in Fig. S18, HCO_3^- (1417 cm⁻¹), $b-CO_3^{2-}$ (1508 cm⁻¹), $m-CO_3^{2-}$ (1555 cm⁻¹), and CO_2^- (1686 cm⁻¹) are detected on the surface of CA/ Ni₁Cu_{NP} N-C after continuous purging of CO₂ and H₂O for 10 min under dark conditions. [56] These carbonates, bicarbonates, and CO2 reactive molecules indicate that the samples have good chemisorption properties for CO2. Under continuous illumination, the key intermediates *COOH and *CO for the generation of CO are detected at 1328, 1638, and 1923 cm⁻¹. *CHO, *CH₂O, and *CH₃O at 1136, 1197, and 1434 cm⁻¹ and several stretching vibrations belonging to the C-H bonds (CH2 and CH₃) at 2800-3100 cm⁻¹ are the critical intermediates for the generation of CH_4 . [57] In particular, *COCO located at 1913 cm^{-1} is the product of *CO coupling and likewise an essential precursor for the generation of C₂ products [58].

Based on the above analysis, we proposed a possible pathway for $\rm CO_2$ reduction. Firstly, $\rm CO_2$ dissolved in aqueous solution formed carbonates and bicarbonates or directly adsorbed on the catalyst surface and formed

 CO_2^- species, which subsequently generated *COOH and *CO through hydrogenation reactions. Due to the good adsorption characteristics of Cu nanoparticles and dispersed Ni single-atom sites for *CO, only a tiny portion of *CO was desorbed to form CO gaseous products. Most adsorbed state *CO will continue participating in the reduction reaction on the catalyst surface to generate CH_4 and C_2H_6 ultimately. The detection of *CHO, *CH2O, *CH3O, *CH2, and *CH3 species on the CA/Ni1CuNP N-C surfaces implies that CH_4 is generated via the formaldehyde pathway. For the evolutionary pathway of the C_2 product, the *CO intermediate would generate the *COCO key intermediate via the C-C coupling, followed by a multi-step hydrogenation reaction to generate C_2H_6 .

To further validate the charge transfer mechanism on CA/Ni₁Cu_{NP} N-C and the synergistic effect between Cu nanoparticles and Ni single-atom sites, theoretical calculations based on density functional theory (DFT) were performed (Fig. S19). As shown in Fig. 9a-b, the work functions of CA and Ni_1Cu_{NP} N-C are 1.908 and 3.359 eV, respectively, which implies that the electrons located in CA will be transferred to Ni₁Cu_{NP} N-C driven by the potential difference until a new stable Fermi energy level is established. The hybridization of CA and Ni₁Cu_{NP} N-C can improve the carrier separation efficiency on CA and transport enough electrons to Ni₁Cu_{NP} N-C to promote CO₂ catalytic reduction. Then, the electron difference densities (EDD) of CA/Ni₁ N-C and CA/Ni₁Cu_{NP} N-C were calculated to further explore the charge separation mechanism. As shown in Fig. 9c-d, the charge at the heterojunction interface is depleted on the CA (green) surface and accumulated on the Ni₁ N-C/Ni₁Cu_{NP} N-C layer (blue). In particular, this charge transfer behavior is enhanced when Cu nanoparticles are introduced, suggesting that it can promote charge transfer and aggregation. Fig. S20 and Fig. 9e-f show the charge density distributions on the Ni₁ N-C and Ni₁Cu_{NP} N-C cut layers, and it can be seen that the Ni centers have high charge densities, which facilitates the adsorption of the reactants/intermediates and the catalytic reactions with multi-electron participation. In the CA/Ni₁Cu_{NP} N-C system, the charge density around the electron-rich center Ni-N₄ shows an asymmetric distribution in the presence of Cu nanoparticles. This asymmetric active site is favorable for coupling adjacent *CO, thus promoting the generation of C₂ products. [59,60] Finally, we calculated the density of states (DOS) of CA/Ni₁ N-C and CA/Ni₁Cu_{NP} N-C. As shown in Fig. S21, CA/Ni₁Cu_{NP} N-C has a more vital and rightward-shifted DOS, which allows for stronger binding of reactants/intermediates on its surface [15].

3. Conclusion

In conclusion, we anchored Ni single-atom and Cu nanoparticle sites on nitrogen-doped carbon via a facile ion-exchange strategy and calcination, and subsequently hybridized it with CoAl-LDH based on ultrasonic self-assembly. The intermediates of the photocatalytic reactions were determined by in situ characterization techniques to speculate the possible ${\rm CO}_2$ evolution pathways. The charge transfer mechanism on the

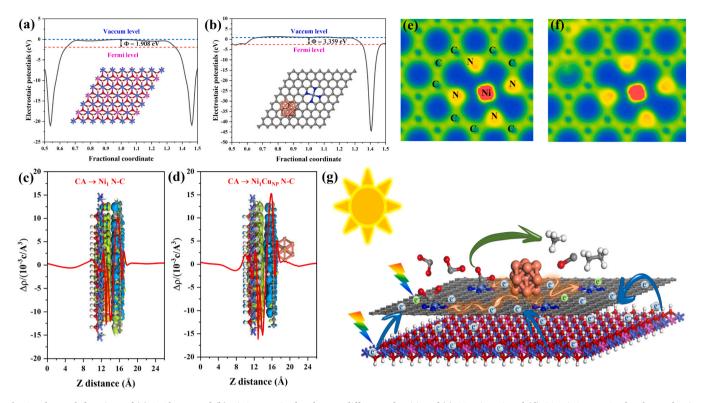


Fig. 9. The work functions of (a) CoAl-LDH and (b) Ni_1Cu_{NP} N-C. The electron difference densities of (c) CA/Ni_1 N-C and (d) CA/Ni_1Cu_{NP} N-C. The charge density distributions on the (e) Ni_1 N-C cut layers of CA/Ni_1 N-C and (f) Ni_1Cu_{NP} N-C cut layers of CA/Ni_1Cu_{NP} N-C. (g) The possible photocatalytic CO_2 reduction mechanism on CA/Ni_1Cu_{NP} N-C.

CA/Ni₁Cu_{NP} N-C and the potential synergistic interactions between Ni single atoms and Cu nanoparticles were explored in depth by DFT calculations. As shown in Fig. 9g, the photogenerated electrons located on CoAl-LDH will be transferred to N-C under the effect of the internal potential and ultimately gathered on the dispersed Ni single-atom sites, providing sufficient power for CO₂ reduction. The loading of Cu nanoparticles can significantly promote charge transfer. In addition, it can change the charge density distribution on N-C to create asymmetric Ni-N₄ sites, enhancing key intermediates' adsorption. As a result, the yields of CH₄ and C₂H₆ for the best samples were 35.245 and 25.328 μ mol g⁻¹ h⁻¹ with electron selectivity of 40.24% and 50.61%, respectively.

3.1. Experimental procedures

Detailed descriptions of the preparation processes, characterization techniques, photoelectrochemical tests, photocatalytic $\rm CO_2$ reduction performance tests, and DFT calculation setup are available in the Supplementary Information.

CRediT authorship contribution statement

Wei-guo Pan: Writing – review & editing, Investigation, Supervision. Chu-fan Li: Investigation, Writing – original draft. Zhen-rui Zhang: Writing – review & editing. Tong Wu: Writing – review & editing. Rui-tang Guo: Conceptualization, Writing – review & editing, Supervision.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.apcatb.2023.123492.

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